DEFECTS IN SOLIDS

Richard J. D. Tilley Emeritus Professor, University of Cardiff, Wales



DEFECTS IN SOLIDS

DEFECTS IN SOLIDS

Richard J. D. Tilley Emeritus Professor, University of Cardiff, Wales



Copyright © 2008 by John Wiley & Sons, Inc. All rights reserved.

Published by John Wiley & Sons, Inc., Hoboken, New Jersey Published simultaneously in Canada

No part of this publication may be reproduced, stored in a retrieval system, or transmitted in any form or by any means, electronic, mechanical, photocopying, recording, scanning, or otherwise, except as permitted under Sections 107 or 108 of the 1976 United States Copyright Act, without either the prior written permission of the Publisher, or authorization through payment of the appropriate per-copy fee to the Copyright Clearance Center, Inc., 222 Rosewood Drive, Danvers, MA 01923, (978) 750-8400, fax (978) 750-4470, or on the web at www.copyright.com. Requests to the Publisher for permission should be addressed to the Permissions Department, John Wiley & Sons, Inc., 111 River Street, Hoboken, NJ 07030, (201) 748-6011, fax (201) 748-6008, or online at http://www.wiley.com/go/permission.

Limit of Liability/Disclaimer of Warranty: While the publisher and author have used their best efforts in preparing this book, they make no representations or warranties with respect to the accuracy or completeness of the contents of this book and specifically disclaim any implied warranties of merchantability or fitness for a particular purpose. No warranty may be created or extended by sales representatives or written sales materials. The advice and strategies contained herein may not be suitable for your situation. You should consult with a professional where appropriate. Neither the publisher nor author shall be liable for any loss of profit or any other commercial damages, including but not limited to special, incidental, consequential, or other damages.

For general information on our other products and services or for technical support, please contact our Customer Care Department within the United States at (800) 762-2974, outside the United States at (317) 572-3993 or fax (317) 572-4002.

Wiley also publishes its books in variety of electronic formats. Some content that appears in print may not be available in electronic format. For more information about Wiley products, visit our web site at www.wiley.com.

Library of Congress Cataloging-in-Publication Data:

Tilley, R. J. D.
Defects in solids/Richard J. D. Tilley.
p. cm.
Includes indexes.
ISBN 978-0-470-07794-8 (cloth)
1. Solids—Defects. 2. Solids—Electric properties. 3. Solids—Magnetic properties.
4. Solids—Optical properties. I. Title.
QC176.8.D44T55 2008
620.1'1—dc22
2008001362

Printed in the United States of America

10 9 8 7 6 5 4 3 2 1

To Professor F. S. Stone who engendered a unique research group spirit and my contemporaries in the University of Bristol, who augmented it.

CONTENTS

Preface		XV
1. Poin	t Defects	1
1.1	Introduction	1
1.2	Point and Electronic Defects in Crystalline Solids	4
1.3	Electronic Properties: Doped Silicon and Germanium as Examples	7
1.4	Optical Properties: F Centers and Ruby as Examples	10
1.5	Bulk Properties	
	1.5.1 Unit Cell Dimensions	13
	1.5.2 Density	14
	1.5.3 Volume	16
	1.5.4 Young's Modulus (the Elastic Modulus)	17
1.6	1.6 Thermoelectric Properties: The Seebeck Coefficient as an Example	
1.7	Point Defect Notation	20
1.8	Charges on Defects	
	1.8.1 Electrons and Electron Holes	21
	1.8.2 Atomic and Ionic Defects	22
1.9	Balanced Populations of Point Defects: Schottky and Frenkel Defects	25
	1.9.1 Schottky Defects	25
	1.9.2 Frenkel Defects	27
1.10	Antisite Defects	
1.11	Defect Formation and Reaction Equations	31
	1.11.1 Addition and Subtraction of Atoms	31
	1.11.2 Equation Formalism	31
	1.11.3 Formation of Antisite Defects	33
	1.11.4 Nickel Oxide	33
	1.11.5 Cadmin Oxide	36
	1.11.7 Ternary Oxides	37
1.12	Combinations of Point Defects in Pure Materials	38
1.13	Structural Consequences of Point Defect Populations	39
1.14	Answers to Introductory Questions	40

	Problems and Exercises			
	References			
	Furth	er Reading	44	
2.	Intri	ssic Point Defects in Stoichiometric Compounds	45	
	2.1	Equilibrium Population of Vacancies in a Monatomic Crystal	45	
	2.2	Equilibrium Population of Self-Interstitials in a Monatomic Crystal	50	
	2.3	Equilibrium Population of Schottky Defects in a Crystal	52	
	2.4	Lithium Iodide Battery	54	
	2.5	Equilibrium Population of Frenkel Defects in a Crystal	56	
	2.6	Photographic Film	58	
	2.7	Photochromic Glasses	62	
	2.8	Equilibrium Population of Antisite Defects in a Crystal	64	
	2.9	Intrinsic Defects: Trends and Further Considerations	65	
	2.10	Computation of Defect Energies	66	
		2.10.1 Defect Calculations	66	
		2.10.2 Point Defect Interactions	68	
		2.10.3 Atomistic Simulation	70	
		2.10.4 The Shell Model 2.10.5 Defect Formation Energy	72	
		2.10.6 Quantum Mechanical Calculations	76	
	2.11	Answers to Introductory Questions	77	
	Proble	ems and Exercises	78	
	Refer	ences	81	
	Furth	er Reading	81	
3.	Exter	ded Defects	83	
	3.1	Dislocations	83	
	3.2	Edge Dislocations	85	
	3.3	Screw Dislocations	90	
	3.4	Mixed Dislocations	93	
	3.5	Unit and Partial Dislocations	94	
	3.6	Multiplication of Dislocations	99	
	3.7	Interaction of Dislocations and Point Defects	99	
		3.7.1 Dislocation Loops	99	
		3.7.2 Dislocation Climb	102	
	2.0		103	
	3.8	5 Dislocations in Nonmetallic Crystals		
	3.9	Internal Boundaries	107	
	5.10 2.11	Low-Angle Grain Boundaries		
	3.11	I win Boundaries	110	
	3.12	Antipnase Boundaries	114	

3.13	Doma	116	
	3.13.1	Magnetic Structures	116
	3.13.2	Ferroelectric Structures	118
	3.13.3	Ferroic Structures	119
3.14	Exterr	120	
	3.14.1	Optical Characteristics of Polycrystalline Solids	122
	3.14.2	Electronic Properties of Interfaces	122
	3.14.3	Varistors	124
	3.14.4	Positive Temperature Coefficient Thermistors	126
3.15	Volun	ne Defects and Precipitates	128
3.16	Answers to Introductory Questions		129
Probl	lems and	131	
Furth	er Read	ling	133
4. Stru	ctural A	Aspects of Composition Variation	135
4 1	4.1 Composition Variation and Nonsteichiometry		
7.1	4 1 1	Dhase Diagrams and the Convictance of Solids	135
	4.1.1	Nonstoichiometric Compounds	133
	4.1.3	Phase Diagrams and Composition	137
4.2	Substitutional Solid Solutions Point Defects and Departures from Stoichiometry		140
4.3			143
	4.3.1	Substitution: Gallium Arsenide, GaAs	143
	4.3.2	Vacancies: Cobalt Oxide, CoO	145
	4.3.3	Interstitials: La ₂ CuO ₄ and Sr ₂ CuO ₂ F ₂	145
	4.3.4	Interstitial Impurities: Alloys and Hydrides	147
	4.3.5	Defect Variation: Zinc Oxide, ZnO	147
	4.3.6	Defect Spinels, M_2O_3	148
4.4	Defec	t Clusters	149
	4.4.1	Point Defect Aggregations	149
	4.4.2	Iron Oxide, Wüstite	150
	4.4.3	Hyperstoichiometric Uranium Dioxide	152
	4.4.4	Anion-Excess Fluorite Structures	155
	4.4.5	Amon-Dencient Fluorite Structures	157
4.5	Interpolation Intercalation		161
4.6			165
4.7	Linear	Linear Defects	
4.8	Modu	lar Structures	170
	4.8.1	Crystallographic Shear Phases	171
	4.ð.2	r winning and Composition Variation	1/6
	4.0.3 484	Perovskite-Related Structures	170
	4.8.5	Misfit-Layer Compounds	184
	4.8.6	Niobium Pentoxide Block Structures	187

X CONTENTS

4.9	Ordering and Assimilation	190
4.10	Modulated Structures	192
	4.10.1 Structure Modulations	192
	4.10.2 Yttrium Oxyfluoride and Related Structures	193
	4.10.3 U ₃ O ₈ -Related Structures	195
	4.10.4 Sr_xTiS_3 and Related Structures	196
4.11	Answers to Introductory Questions	198
Probl	ems and Exercises	199
Furth	er Reading	202
5. Defe	ets and Diffusion	205
5.1	Diffusion	205
5.2	Diffusion in Solids	207
5.3	Random-Walk Diffusion in Crystals	213
5.4	Diffusion Mechanisms	216
	5.4.1 Vacancy Diffusion	216
	5.4.2 Interstitial Diffusion	217
	5.4.3 Interstitialcy Diffusion	217
	5.4.4 Impurity and Cluster Diffusion Mechanisms	219
	5.4.5 Dillusion Pains	221
5.5	Point Detect Concentration and Diffusion	227
5.6	Correlation Factors	227
5.7	Temperature Variation of the Diffusion Coefficient	231
5.8	Temperature Variation and Intrinsic Diffusion	236
5.9	Diffusion Mechanisms and Impurities	240
5.10	Chemical and Ambipolar Diffusion	241
5.11	Dislocation and Grain Boundary Diffusion	243
5.12	Diffusion in Amorphous and Glassy Solids	245
5.13	Answers to Introductory Questions	245
Probl	ems and Exercises	246
Furth	er Keading	250
6. Intri	nsic and Extrinsic Defects in Insulators: Ionic Conductivity	251
6.1	Ionic Conductivity	251
	6.1.1 Ionic Conductivity in Solids	251
	6.1.2 Fundamental Concepts	252
6.2	Mechanisms of Ionic Conductivity	255
	6.2.1 Random-Walk Model	255
	6.2.2 Relationship between Ionic Conductivity and Diffusion Coefficient	259
	6.2.3 Ionic Conductivity and Defects	261
6.3	Impedance Measurements	263
6.4	Electrochemical Cells and Batteries	265

6.5	Disor	dered Cation Compounds	268
6.6	β-Alu	mina Oxides	270
	6.6.1 6.6.2	Idealized Structures of the β -Alumina-Related Phases Defects in β -Alumina	270 272
	6.6.3	Defects in β'' -Alumina	274
	0.0.4	Batteries Using B"-Alumina	274
67	Enhar	accement of Ionic Conductivity	278
6.8	Calcia	a-Stabilized Zirconia and Related Fast Oxygen Ion Conductors	280
	6.8.1	Structure and Oxygen Diffusion in Fluorite Structure Oxides	280
	6.8.2	Free Energy and Stoichiometry of Oxides	281
	6.8.3	Oxygen Sensors	282
	6.8.4	Oxygen Pumps and Coulometric Titrations	284
6.9	Protor	n (H ⁺ Ion) Conductors	285
6.10	Solid	Oxide Fuel Cells	287
6.11	Answ	ers to Introductory Questions	288
Probl	ems and	d Exercises	290
Furth	er Read	ling	294
7. Nons	toichio	metry and Intrinsic Electronic Conductivity	297
7.1	Nonst	oichiometry and Electronic Defects in Oxides	297
	7.1.1	Electronic and Ionic Compensation	297
	7.1.2	Metal-Excess Phases	298
	7.1.3	Oxygen-Excess Phases	299
7.2	Condu	activity and Defects	301
	7.2.1	Conductivity and Defect Concentrations	301
	7.2.2	Holes, Electrons, and Valence	302
	7.2.3	Localized Electrons and Polarons	303
	7.2.4	Detects and Hopping Conductivity	304
	7.2.6	Seebeck Coefficient and Stoichiometry	300
73	Stoich	nometry Defect Populations and Partial Pressures	309
7.5	731	Fauilibrium Partial Pressures	309
	7.3.2	Phase Rule	312
7.4	Variat	ion of Defect Populations with Partial Pressure	315
	7.4.1	Metal-Excess Oxides	315
	7.4.2	Oxygen-Excess Oxides	317
	7.4.3	Ba ₂ In ₂ O ₅	318
7.5	Brouv	ver Diagrams	320
	7.5.1	Initial Assumptions	320
	7.5.2	Defect Equilibria	321
	7.5.3	Stoichiometric Point: Ionic Defects	322

	7.5.4 7.5.5 7.5.6 7.5.7	Near Stoichiometry: Ionic Defects High X ₂ Partial Pressures: Ionic Defects Low X ₂ Partial Pressures: Ionic Defects Complete Diagram: Ionic Defects	324 326 327 328
7.6	Brouv	wer Diagrams: Electronic Defects	329
	7.6.1 7.6.2 7.6.3 7.6.4 7.6.5	Electronic Defects Near Stoichiometry: Electronic Defects High X ₂ Partial Pressures: Electronic Defects Low X ₂ Partial Pressures: Electronic Defects Complete Diagram: Electronic Defects	329 330 333 334 335
7.7	Brouv	wer Diagrams: More Complex Examples	336
	7.7.1 7.7.2	Cr ₂ O ₃ Ba ₂ In ₂ O ₅	336 338
7.8	Brouv	wer Diagrams: Effects of Temperature	339
7.9	Polyn	nomial Forms for Brouwer Diagrams	340
	7.9.1 7.9.2	Ionic Defects Electronic Defects	340 343
7.10	Answ	vers to Introductory Questions	344
Probl	blems and Exercises		345
Refei	References		
Furth	er Read	ling	349
8. Nons	toichio	metry and Extrinsic Electronic Conductivity	351
8.1	Effect	351	
8.2	Impur	rities in Oxides	352
	8.2.1 8.2.2 8.2.3 8.2.4	Donor Doping Donor Doping of Cr ₂ O ₃ Acceptor Doping Acceptor Doping of NiO	352 352 353 354
8.3	Negat	tive Temperature Coefficient (NTC) Thermistors	355
8.4	Brouv	wer Diagrams for Doped Systems	358
	8.4.1 8.4.2	Construction General Trends: MX	358 360
8.5	Metal	s and Insulators	363
	8.5.1 8.5.2 8.5.3	Acceptor Doping into La_2CuO_4 Donor Doping into Nd_2CuO_4 Spinel System $Li_{1+x}Ti_{2-x}O_4$	363 365 366
8.6	Cupra	ate High-Temperature Superconductors	367
	8.6.1 8.6.2 8.6.3	Perovskite-Related Structures and Series Hole Doping Defect Structures	367 370 373

8.7	Mixed Electronic/Ionic Conductors	378
	8.7.1 Fluorite Structure Oxides: $CeO_{2-\delta}$	378
	8.7.2 Layered Structures: $Li_x MX_2$	380
	8.7.3 Acceptor Doping in Perovskite Structure Oxides	381
8.8	Mixed Proton/Electronic Conductors	385
	8.8.1 Proton Mixed Conductors	385
	8.8.2 Brouwer Diagram Representation of Mixed Proton Conductivity	387
	8.8.3 Charge Carrier Map Representation of Mixed Conductivity	389
8.9	Choice of Compensation Mechanism	391
8.10	Answers to Introductory Questions	392
Prob	ems and Exercises	394
Furth	er Reading	397
9. Mag	netic and Optical Defects	399
9.1	Magnetic Defects	399
9.2	Magnetic Defects in Semiconductors	403
9.3	Magnetic Defects in Ferrites	405
9.4	Charge and Spin States in Cobaltites and Manganites	406
9.5	Extended Magnetic Defects	
9.6	Optical Defects	
	9.6.1 Absorption and Scattering	412
	9.6.2 Energy Levels	413
	9.6.3 Energy Levels in Solids	416
9.7	Pigments, Minerals and Gemstones	417
	9.7.1 Transition-Metal and Lanthanide Ion Colors	417
	9.7.2 Colors and Impurity Defects	418
9.8	Photoluminescence	420
	9.8.1 Energy Degradation and Down-conversion	420
	9.8.2 Up-conversion	421
9.9	Solid-State Lasers	428
	9.9.1 Emission of Radiation	428
	9.9.2 Ruby Laser: Three-Level Lasers	429
	9.9.3 Neodymium (Nd ³⁺) Laser: Four-Level Lasers	430
9.10	Color Centers	
	9.10.1 The F Center	432
	9.10.2 Electron and Hole Centers	433
	9.10.3 Surface Color Centers: 0.10.4 Complex Color Centers: Laser and Photonic Action	435
0.11	2.10.4 Complex Color Centers. Laser and Filotomic Action	430
9.11	Electrochromic Films	438
9.12	Photoinduced Magnetism	440
9.13	Answers to Introductory Questions	441

F	Problem	ns and Exercises	443
F	Further Reading		
Sup	Supplementary Material		
S 1	Crystal Structures		
	S1.1	Crystal Systems and Unit Cells	447
	S1.2	Crystal Planes and Miller Indices	448
	S1.3	Directions	451
	S1.4	Crystal Structures	452
	Furth	er Reading	460
S 2	Band Theory		
	S2.1	Energy Bands	461
	S2.2	Insulators, Semiconductors and Metals	461
	S2.3	Point Defects and Energy Bands in Semiconductors and Insulators	463
	S2.4	Transition-Metal Oxides	464
S 3	Seebeck Coefficient		
	S3.1	Seebeck Coefficient and Entropy	465
	S3.2	Seebeck Coefficient and Defect Populations	468
S 4	Schottky and Frenkel Defects		
	S4.1	Equilibrium Concentration of Schottky Defects Derived	
		from Configurational Entropy	470
	S4.2	Stirling's Approximation	473
	S4.3	Equilibrium Concentration of Frenkel Defects Derived	
		from Configurational Entropy	474
S5	Diffu	sion	476
	\$5.1	Diffusion Equations	476
	S5.2	Non-Steady-State Diffusion	476
	S5.3	Random-Walk Diffusion	478
	S5.4	Concentration Profile	484
	\$5.5	Fick's Laws and the Diffusion Equations	485
	\$5.6	Penetration Depth	489
S 6	Magr	netic Properties	489
	S6.1	Atomic Magnetism	489
	S6.2	Types of Magnetic Material	490
	S6.3	Crystal Field Splitting	492
Ans	swers t	o Problems and Exercises	495
For	mula]	Index	507
Subject Index		513	

During the course of the last century, it was realized that many properties of solids are controlled not so much by the chemical composition or the chemical bonds linking the constituent atoms in the crystal but by faults or defects in the structure. Over the course of time the subject has, if anything, increased in importance. Indeed, there is no aspect of the physics and chemistry of solids that is not decisively influenced by the defects that occur in the material under consideration. The whole of the modern silicon-based computer industry is founded upon the introduction of precise amounts of specific impurities into extremely pure crystals. Solid-state lasers function because of the activity of impurity atoms. Battery science, solid oxide fuel cells, hydrogen storage, displays, all rest upon an understanding of defects in the solid matrix.

As the way in which defects modify the properties of a solid has been gradually understood, the concept of a defect has undergone considerable evolution. The earliest and perhaps simplest concept of a defect in a solid was that of a wrong atom, or impurity, in place of a normal atom in a crystal—a so-called point defect. Not long after the recognition of point defects, the concept of linear defects, dislocations, was invoked to explain a number of diverse features including the mechanical properties of metals and the growth of crystals. In recent years it has become apparent that defect interactions, aggregation, or clustering is of vital importance.

The advance in understanding of defects has been made hand in hand with the spectacular development of sophisticated experimental techniques. The initial break-through was in X-ray diffraction—a technique that still remains as the foundation of most studies. A further change came with the development of transmission electron microscopy that was able, for the first time, to produce images of defects at an atomic scale of resolution. Since then, advances in computing techniques, together with the availability of powerful graphics, have thrown quite new light on the defect structure of materials.

Besides the multiplicity of defects that can be envisaged, there is also a wide range of solid phases within which such defects can reside. The differences between an alloy, a metallic sulfide, a crystalline fluoride, a silicate glass, or an amorphous polymer are significant. Moreover, developments in crystal growth and the production of nanoparticles have changed the perspective of earlier studies, which were usually made on polycrystalline solids, sometimes with uncertain degrees of impurity present.

All of these changes have meant that the view of defects in solids has changed considerably over the last 20 years or so. This book is aimed at presenting an overview of this information. However, the topic encompasses a huge subject area, and selection has been inevitable. Moreover, courses about defects in solids, whether to undergraduate or postgraduate students in chemistry, physics, geology, materials science, or engineering, are usually constrained to a relatively small part of the curriculum. Nevertheless, the material included in this book has been chosen so that not only basics are covered but also aspects of recent research where exciting frontiers lie. Unfortunately, the influence of defects upon mechanical properties is mostly excluded, and the important area of surfaces and surface defects is only mentioned in passing. Similarly, the area of studies prefixed by nano- has been bypassed. This is because the literature here is growing at an enormous rate, and time was not available to sift through this mountain of data. Although these omissions are regrettable, they leave scope for future volumes in this series. Finally, it must be mentioned that most emphasis has been placed upon principles, leaving little space for description of important experimental details. This is a pity, but within the constraints of time and space, became inevitable. Again, a future volume could correct this shortcoming.

The first four chapters introduce basic concepts that are developed to build up a framework for understanding defect chemistry and physics. Thereafter, chapters focus rather more on properties related to applications. Chapter 5 describes diffusion in solids; Chapter 6, ionic conductivity; Chapters 7 and 8 the important topics of electronic conductivity, both intrinsic (Chapter 7) and extrinsic (Chapter 8). The final chapter gives a selected account of magnetic and optical defects.

To assist in understanding, each chapter has been prefaced with three "introductory questions" that focus the reader upon some of the important points to be raised in the following text. These are answered at the end of each chapter. In addition, end-of-chapter questions aid understanding of the preceding material. The first set of these consists of multiple choice questions—a "quick quiz," to test knowledge of terms and principles. This is followed by a number of more traditional problems and calculations to build skills and understanding in more depth. In addition, supplementary material covering the fundamentals of relevant topics such as crystallography and band theory are included so that, in the first instance, a reader will not have to look elsewhere for this information.

Each chapter contains a short list of additional sources that expand or give a different perspective on the material in the preceding chapter. Most of these are books or original scientific literature, and only a few web sources are listed. The Internet provides a data bank of considerable power but has two drawbacks: It is easy to become swamped by detail (often a simple search will throw up a million or more allegedly relevant pages), and much of the information located has an ephemeral nature. Rather than list a large number of sites, the reader is encouraged to use a search engine and keywords such as "magnetic defects," which will open an alternative perspective on the subject to that presented in this book.

I have been particularly helped in the compilation of this book by family, friends, and colleagues. Professor R. B. King first suggested the project, and Drs. E. E. M. Tyler, G. J. Tilley, and R. D. Tilley made suggestions about contents and scope. Professor F. S. Stone offered encouragement and kindly offered to read sections in

draft form, resulting in invaluable advice and comment that added substantially to the clarity and balance. Mr. A. Coughlin was constantly encouraging and gave assistance when mathematical discussions of topics such as polynomials or random walks became opaque and provided valued help into the proofs. The staff of the Trevithick Library, University of Cardiff, have been helpful at all times and continually located obscure references. The staff at Wiley, gave constant advice, help, and encouragement. Dr. John Hutchison, University of Oxford, provided stunning micrographs for which I am greatly indebted. Finally, my gratitude to my wife Anne cannot be understated. Her tolerance of my neglect and her continued encouragement has allowed this project to reach a conclusion.

Comments and queries will be gratefully received and can be sent to tilleyrj@cardiff.ac.uk or rjdtilley@yahoo.co.uk.

R. J. D. TILLEY

Point Defects

What is a point defect? What is the "effective charge" on a defect? What is an antisite defect?

1.1 INTRODUCTION

Defects play an important part in both the chemical and physical behavior of solids, and much of modern science and technology centers upon the exploitation or suppression of the properties that defects confer upon a solid. Batteries, fuel cells, displays, data storage, and computer memories all directly utilize, or have evolved from, an understanding and manipulation of defects in inorganic materials. This technology has been developed over some 80-100 years and started with the simplest concepts. However, as the effect of defects upon the properties of the solid gradually became appreciated, the concept of a defect has undergone considerable evolution. The simplest notion of a defect in a solid was the idea of a mistake such as a missing atom or an impurity in place of a normal atom. These structurally simple defects are called point defects. Not long after the recognition of point defects, the concept of more complex structural defects, such as linear defects termed dislocations, was invoked to explain the mechanical properties of metals. In the same period it became apparent that planar defects, including surfaces and grain boundaries, and volume defects such as rods, tubes, or precipitates, have important roles to play in influencing the physical and chemical properties of a solid.

Defects can thus be arranged in a dimensional hierarchy (Fig. 1.1a-1.1d):

- 1. Zero-dimensional defects-point defects
- 2. One-dimensional (linear) defects-dislocations
- 3. Two-dimensional (planar) defects-external and internal surfaces
- 4. Three-dimensional (volume) defects—point defect clusters, voids, precipitates.

This and the following chapter are concerned with point defects.

Defects in Solids, by Richard J. D. Tilley

Copyright © 2008 John Wiley & Sons, Inc.



Figure 1.1 Defects in crystalline solids: (*a*) point defects (interstitials); (*b*) a linear defect (edge dislocation); (*c*) a planar defect (antiphase boundary); (*d*) a volume defect (precipitate); (*e*) unit cell (filled) of a structure containing point defects (vacancies); and (*f*) unit cell (filled) of a defect-free structure containing "ordered vacancies."

In addition to the defects listed above, which may be termed structural defects, there are also electronic defects. The first of these are electrons that are in excess of those required for chemical bonding and that, in certain circumstances, constitute charged defects that can carry current. In addition, current in some materials is carried by particles



that behave rather like positive electrons, and these form the second type of electronic defect.¹ They are called electron holes, positive holes, or more often just holes.

Besides the multiplicity of defects that can be envisaged, there is a wide range of host solid phases within which such defects can reside. The differences between an alloy, a metallic sulfide, a crystalline fluoride, or a silicate glass are significant from

¹These are not the positive equivalent of an electron, a positron, because such a particle would be eliminated instantaneously by combination with an ordinary electron, but are virtual particles equivalent to the absence of an electron. They can be considered to be analogs of a vacancy, which is the absence of an atom.

both a chemical or physical viewpoint. By default, defects have come to be associated with crystalline solids because a crystal has a regular repetition of atoms throughout its volume.² A disturbance of this regularity then constitutes the defect (Fig. 1.1a-1.1d). In this sense, the nature of a defect in, for example, a glass or amorphous polymer, is more difficult to picture.

When circumstances permit, defects can order. This may happen at low temperatures, for example, when defect interactions overcome the disordering effect due to temperature. From a diffraction point of view the ordered sample no longer contains defects, as the characteristic signs disappear. For example, in X-ray analysis broadened diffraction spots and diffuse scattering, both characteristic of disordered crystals, disappear, to be replaced by additional sharp reflections. In such cases the defects are incorporated or assimilated into the crystal structure and so effectively vanish. The original unit cell of the defect-containing phase is replaced by a new unit cell in which the original "defects" are now integral components of the structure (Fig. 1.1e and 1.1f).

1.2 POINT AND ELECTRONIC DEFECTS IN CRYSTALLINE SOLIDS

The simplest localized defect in a crystal is a mistake at a single atom site in a pure monatomic crystal, such as silicon or iron. Such a defect is called a point defect. Two different types of simple point defect can occur in a pure crystal of an element, M. An atom can be imagined to be absent from a normally occupied position, to leave a vacancy, given the symbol V_M (Fig. 1.2). This may occur, for instance, during crystal growth. A second defect can also be envisaged; namely an extra atom incorporated into the structure, again, say, during crystallization. This extra atom is forced to take up a position in the crystal that is not a normally occupied site: it is called an interstitial site, and the atom is called an interstitial atom (or more often simply an interstitial), given the symbol M_i , (Fig. 1.2). If it is necessary to stress that the interstitial atom is the same as the normal atoms in the structure, it is called a self-interstitial atom.

The various point defects present in a crystal as grown, for example, vacancies, interstitials, excess electrons, holes, and other arrangements, are called *native defects*. These native defects can arise in several ways. They can be introduced during crystal growth, as mentioned above, especially if this is rapid and crystallization is far removed from equilibrium. Defects can also form after the crystal is developed if the crystal is subjected to irradiation by high-energy particles or other forms of radiation. Defects that result from such processes are called *induced defects*. When a pure crystal is heated at a moderate temperature for a long period, a process called annealing, the number of native defects will gradually change. However, no matter how long the sample is annealed, a population of point defects will always remain, even in the purest crystal. These point defects are in thermodynamic equilibrium

²An introduction to crystal structures and nomenclature is given in the Supplementary Material Section S1.



Figure 1.2 Point defects in a pure monatomic crystal of an element M, a vacancy, V_M , and a self-interstitial, M_i .

(Chapter 2) and cannot be eliminated from the solid. They are called *intrinsic* point defects. This residual population is also temperature dependent, and, as treated later (Chapter 2), heating at progressively higher temperature increases the number of defects present.

As well as these intrinsic structural defect populations, electronic defects (excess electrons and holes) will always be found. These are also intrinsic defects and are present even in the purest material. When the equilibria among defects are considered, it is necessary to include both structural and electronic defects.

Turning to pure *compounds*, such as CaO, MgAl₂O₄, or FeS, the same intrinsic defects as described above can occur, but in these cases there is more than one set of atoms that can be affected. For example, in a crystal of formula MX, vacancies might occur on metal atom positions, written V_M , or on nonmetal atom positions, given the symbol V_X , or both. Similarly, it is possible to imagine that interstitial metal atoms, written M_i , or nonmetal atoms, written X_i , might occur (Fig. 1.3). The different sets of atom types are frequently called a sublattice, so that one might speak of vacancies on the metal sublattice or on the nonmetal sublattice.

No material is completely pure, and some foreign atoms will invariably be present. If these are undesirable or accidental, they are termed impurities, but if they have been added deliberately, to change the properties of the material on purpose, they are called dopant atoms. Impurities can form point defects when present in low concentrations, the simplest of which are analogs of vacancies and interstitials. For example, an impurity atom A in a crystal of a metal M can occupy atom sites normally occupied by the parent atoms, to form substitutional point defects, written A_M , or can occupy interstitial sites, to form interstitial point defects, written A_i (Fig. 1.4). The doping of aluminum into silicon creates substitutional point defects as the aluminum atoms occupy sites normally filled by silicon atoms. In compounds, the impurities can affect one or all sublattices. For instance, natural sodium chloride often contains



Figure 1.3 Point defects in a crystal of a pure compound, MX, V_M , a metal vacancy; V_X a nonmetal vacancy; M_i , a metal (self-)interstitial; and X_i a nonmetal (self-)interstitial.

potassium impurities as substitutional defects on sites normally occupied by sodium ions, written K_{Na} , that is, the impurities are associated with the metal sublattice. Impurities are called extrinsic defects. In principle, extrinsic defects can be removed by careful processing, but in practice this is very difficult to achieve completely.

Impurities can carry a charge relative to the host structure, as, for example, with a Ca^{2+} ion substituted on a Na^+ site in NaCl or F^- substituted for O^{2-} in CaO. In essence, this means that the impurity carries a different chemical valence, that is,



Figure 1.4 Impurity or dopant (A) point defects in a crystal of material M, substitutional, A_M ; interstitial, A_i .

it comes from a different group of the periodic table than the host atoms. Deliberate introduction of such impurities is called aliovalent or altervalent doping. The introduction of charged impurities will upset the charge neutrality of the solid, and this must be balanced in some way so as to restore the electronic neutrality of the crystal. One way in which this compensation can be achieved is by the incorporation of other vacancies or interstitial atoms that carry a balancing charge. The substitution of a Ca²⁺ ion for Na⁺ in NaCl can be balanced, for example, by the introduction at the same time, of a sodium ion vacancy, V_{Na} , at the same time. This mechanism is called ionic compensation, structural compensation, or less often self-compensation. However, compensation can also take place electronically by the introduction of appropriate numbers of electrons or holes. For example, the substitution of F⁻ for $O^{2^{-}}$ in CaO could be balanced by the introduction of an equivalent number of electrons, one per added F⁻. Impurities that are compensated by excess electrons in this way are called donors or donor impurities, while those that are compensated by addition of holes are called acceptors or acceptor impurities. One consequence of these alternatives is that the electrons and holes present in the solid must be included in the overall accounting system used for assessing the defects present in a solid.

The importance of point defects in a crystal cannot be overstated. They can change the physical properties of a solid significantly. To introduce the range of changes possible, Sections 1.3-1.6 outline some of the physical properties that are influenced in this way.

1.3 ELECTRONIC PROPERTIES: DOPED SILICON AND GERMANIUM AS EXAMPLES

Silicon, Si, lies at the heart of most current electronic devices. Both silicon and the similar semiconductor element germanium, Ge, crystallize with the diamond structure (Fig. 1.5). In this structure each atom is surrounded by four others arranged at the corners of a tetrahedron. Each atom has four outer electrons available for chemical bonding, and these are completely taken up by creating a network of tetrahedrally oriented sp^3 -hybrid bonds. Although pure silicon and germanium are intrinsic semiconductors, these properties are inadequate for the creation of sophisticated electronic materials for which selected impurity doping is necessary. The resulting materials are called extrinsic semiconductors because of the extrinsic nature of the defects that give rise to the important conductivity changes.

Impurity atoms will upset the orderly arrangement of bonding electrons and this changes the electronic properties. Doping with a very small amount of an impurity from the next *higher* neighboring group of the periodic table, phosphorus (P), arsenic (As), or antimony (Sb), results in the formation of substitutional point defects in which the impurities occupy normal sites, for example, a phosphorus atom on a site normally occupied by a silicon atom, P_{Si} (Fig. 1.6*a*). Each of these atoms has five valence electrons available for bonding, and after using four to form the four *sp*³-hybrid bonds, one electron per impurity atom is left over. These electrons are easily liberated from the impurity atoms by thermal energy and are



Figure 1.5 Diamond structure: (a) unit cell and (b) viewed with [111] vertical.

then (more or less) free to move through the crystal under the influence of an external electric field. Each dopant atom contributes one extra electron to the crystal. They are defects, and are given the symbol e' (the superscript indicates a single negative charge relative to the surroundings) to differentiate them from ordinary electrons. The doped material conducts mainly using these electrons, and, as they are negatively charged, the solid is called an *n*-type semiconductor. The atoms P, As, or Sb in silicon or germanium are called donors as they donate extra electrons to the crystal. In terms of band theory they are said to occupy states in the conduction band (Supplementary Material S2), the donors themselves being represented by localized energy levels just below the conduction band in the crystal (Fig. 1.6b).

An analogous situation arises on doping silicon or germanium with elements from the next *lower* neighboring periodic table group, aluminum (Al), gallium (Ga), and indium (In). The impurity atoms again form substitutional defects, such as Al_{Si} (Fig. 1.6*c*). In this case the impurities have only three outer bonding electrons available, which are not sufficient to complete four bonds to the surrounding atoms. One bond is an electron short. It simplifies understanding if the missing electron is regarded as a hole, represented by the symbol h[•]. The superscript indicates that the hole carries a



Figure 1.6 (*a*) Donor impurity (P_{Si}) in a silicon crystal. (*b*) Donor energy levels below the conduction band. (*c*) Acceptor impurity (Al_{Si}) in a silicon crystal. (*d*) Acceptor energy levels above the valence band.

positive charge relative to the surroundings. Each impurity atom introduces one positive hole into the array of bonds within the crystal. Thermal energy is sufficient to allow the holes to leave the impurity atom, and these can also move quite freely throughout the crystal. The doped material conducts mainly using these holes, and, as they are positively charged, the solid is called a p-type semiconductor. The impurities are termed acceptors because they can be thought of as accepting electrons from the otherwise full valence band. On an energy band diagram the acceptors are represented by normally vacant energy levels just above the top of the valence band (Fig. 1.6d), which become occupied by electrons from that band on thermal excitation. The holes thereby created in the valence band provide the means for conductivity.

The idea of a hole is widely used throughout electronics, but its exact definition varies with circumstances. It is frequently convenient to think of a hole as a real particle able to move throughout the crystal, a sort of positive electron. It is equally possible to think of a hole as an electron missing from a localized site such as an atom or a covalent bond. Thus a metal *cation* M^{2+} could be regarded as a metal *atom* plus two localized holes (M + 2 h[•]). In band theory, a hole is generally regarded as an electron missing from the top of the valence band. All of these designations are more or less equivalent, and the most convenient of them will be adopted in any particular case.

The unique electronic properties of semiconductor devices arise at the regions where *p*-type and *n*-type materials are in close proximity, as in p-n junctions. Typical impurity levels are about 0.0001 at %, and their inclusion and distribution need to be very strictly controlled during preparation. Without these deliberately introduced point defects, semiconductor devices of the type now commonly available would not be possible.

1.4 OPTICAL PROPERTIES: F CENTERS AND RUBY AS EXAMPLES

Point defects can have a profound effect upon the optical properties of solids. The most important of these in everyday life is color,³ and the transformation of transparent ionic solids into richly colored materials by F centers, described below, provided one of the first demonstrations of the existence of point defects in solids.

Defects can impart color to an otherwise transparent solid if they interact with white light. In general the interaction between a defect and the incident light is via electrons or holes. These may pick up some specific frequency of the incident illumination and in so doing are excited from the low-energy ground state, E_0 , to one or more higher energy excited states E_1 , E_2 , E_3 (Fig. 1.7*a*). The light that leaves the crystal is thus depleted in some frequencies and appears a different color to the incident light. The reverse also happens. When electrons drop from the excited states to the ground state E_0 , they release this energy and the same light frequencies will be emitted (Fig. 1.7*b*). The relationship between the energy gained or lost, ΔE_n , and the frequency, v, or the wavelength, λ , of the light absorbed or emitted, is

$$E_n - E_0 = \Delta E_n = h v_n = \frac{hc}{\lambda_n}$$

where E_n is the energy of the higher energy level (n = 1, 2, 3, ...), h is Planck's constant, and c is the speed of light.

The first experiments that connected color with defects were carried out in the 1920s and 1930s by Pohl, who studied synthetic alkali halide crystals. A number of ways were discovered by which the colorless starting materials could be made to display intense colors. These included irradiation by X rays, electrolysis (with color moving into the crystal from the cathode), or heating the crystals at high temperatures in the vapor of an alkali metal. Pohl was a strict empiricist who did not openly speculate upon the mechanics of color formation, which he simply attributed to the presence of Farbzentren (lit. color centers), later abbreviated to F centers.

Leading theoreticians were, however, attracted to the phenomenon and soon suggested models for F centers. In 1930 Frenkel suggested that an F center was an electron trapped in a distorted region of crystal structure, an idea that was incorrect in this instance but led directly to development of the concepts of excitons and

³Color is the name given to the perception of radiation in the electromagnetic spectrum with a wavelength of between 400 and 700 nm, for an average eye. If all wavelengths in this range are present in a distribution similar to that of radiation from the sun, the light is called white light.